# **Synthesis and Characterization of Bis(3,4-ethylene-dioxythiophene)-(4,4**′**-dialkyl-2,2**′**-bithiazole) Co-oligomers for Electronic Applications**

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The synthesis and characterization of several co-oligomers, 5,5′-bis(4,4′-dialkyl-2,2′ bithiazol-5-yl)-(3,4,3',4'-bis(ethylenedioxy)-2,2'-dithienyl) ( $AT_2A$ ), 5,5'-bis(3,4-ethylenedioxythien-2-yl)-4,4′-dibutyl-2,2′-bithiazole (TBT), and 5,5′-bis(3,4,3′,4′-bis(ethylenedioxy)-2,2′ dithien-5-yl)-4,4′-dibutyl-2,2′-bithiazole (T<sub>2</sub>BT<sub>2</sub>) are described. The solid-state structure of AT2A features *π*-stacking with a short intermolecular distance of 3.5 Å. In thin films, the *<sup>π</sup>*-stacks are parallel to the film substrate as determined by X-ray diffraction. The UV-vis spectra of AT2A films show features that are ascribed to the effects of *π*-stacking. In addition, all these co-oligomers, composed of donor-acceptor moieties, show lowered optical band gaps, mainly because of the low oxidation potential imparted by incorporation of the 3,4 ethylenedioxythiophene (EDOT) moiety. Cyclic voltammetry indicates that these co-oligomers can be either p-doped or n-doped, and the spectroelectrochemical measurements show that the  $AT<sub>2</sub>A$  films are stable upon cycling the oxidation state. All these features make these materials promising for use in electronic devices, e.g., organic thin film transistors.

## **Introduction**

Recently, organic conjugated materials have been studied very intensively as active materials for organic thin film transistors (OTFTs). $1-5$  The unique properties of these organic materials make them more attractive than inorganic semiconductors for applications requiring large area coverage, structural flexibility, or solution processing.6-<sup>8</sup> Currently, outstanding progress has been made in fabricating OTFT devices. $9-12$  The field-effect mobility of some OTFT devices has reached values comparable to those of amorphous Si-TFTs, e.g., singlecrystal pentacene OTFT devices show mobilities of  $1-2.7$  cm<sup>2</sup>V<sup>-1</sup>s<sup>-1</sup> at room temperature,<sup>9,13</sup> a value comparable to that of amorphous Si. These advances in

- (4) Katz, H. E.; Bao, Z. *J. Phys. Chem. B* **2000**, *104*, 671.
- (5) Lovinger, A. J.; Rothberg, L. J. *J. Mater. Res.* **1996**, *11*, 1581. (6) Drury, C. J.; Mutsaers, C. M. J.; Hart, C. M.; Matters, M.; de
- Leeuw, D. M. *Appl. Phys. Lett.* **1998**, *73*, 108. (7) Bao, Z. N.; Rogers, J. A.; Katz, H. E. *J. Mater. Chem.* **1999**, *9*,
- 1895.
- (8) Garnier, F.; Hajlaoui, R.; Yassar, A.; Srivastava, P. *Science* **1994**, *265*, 1684.
- (9) Scho¨n, J. H.; Berg, S.; Kloc, C.; Batlogg, B. *Science* **2000**, *287*, 1022.
- (10) Lin, Y. Y.; Gundlach, D. J.; Nelson, S. F.; Jackson, T. N. *IEEE Trans. Electron Devices* **1997**, *44*, 1325.
- (11) Kagan, C. R.; Mitzi, D. B.; Dimitrakopoulos, C. D. *Science* **1999**, *286*, 945.
- (12) Sirringhaus, H.; Tessler, N.; Friend, R. H. *Science* **1998**, *280*, 1741.
- (13) Lin, Y. Y.; Gundlach, D. J.; Nelson, S. F.; Jackson, T. N. *IEEE Electron Device Lett.* **1997**, *18*, 606.

organic devices suggest that organic conjugated materials may be competitive with traditional Si technology in low-end applications that do not require rapid switching rates. However, some fundamental challenges still remain, and, above all, there remain questions about how charge transport is related to the molecular and solid-state structures of the organic materials. $14-17$ In OTFT devices, early success has been limited to a few types of conjugated materials,  $18$  e.g., thiophene oligomers, polythiophene, pentacene, etc. It is wellknown that both thiophene oligomers and pentacene molecules are packed in the so-called "herringbone" pattern in which intermolecular,  $\pi-\pi$  contacts are minimized.<sup>19-21</sup> Despite the poor  $\pi-\pi$  overlap between neighboring molecules, highly purified pentacene has extremely high charge carrier mobility at low temperature.9,10 There is little evidence to indicate that the herringbone-packing pattern is structurally superior for promoting high carrier mobility to other crystalline prototypes, e.g., *π*-stacks. Therefore, it is important to

- (14) Schön, J. H.; Kloc, C.; Batlogg, B. Phys. Rev. Lett. **2001**, 86, 3843.
- (15) Chung, T. C.; Kaufman, J. H.; Heeger, A. J.; Wudl, F. *Phys. Rev. B* **1984**, *30*, 702.
- (16) Brédas, J. L.; Themans, B.; Andre, J. M.; Chance, R. R.; Silbey, R. *Synth. Met.* **1984**, *9*, 265.
- (17) Hill, M. G.; Mann, K. R.; Miller, L. L.; Penneau, J. F. *J. Am. Chem. Soc.* **1992**, *114*, 2728.
- (18) Katz, H. E.; Bao, Z. N.; Gilat, S. L. *Accounts Chem. Res.* **2001**, *34*, 359.
- (19) Campbell, R. B.; Trotter, J.; Robertson, J. M. *Acta Crystallogr.* **1961**, *14*, 705.
- (20) Horowitz, G.; Bachet, B.; Yassar, A.; Lang, P.; Demanze, F.; Fave, J. L.; Garnier, F. *Chem. Mater.* **1995**, *7*, 1337. (21) Siegrist, T.; Fleming, R. M.; Haddon, R. C.; Laudise, R. A.;
- Lovinger, A. J.; Katz, H. E.; Bridenbaugh, P.; Davis, D. D. *J. Mater. Res.* **1995**, *10*, 2170.

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<sup>(1)</sup> Dimitrakopoulos, C. D.; Malenfant, P. R. L. *Adv. Mater.* **2002**, *14*, 99.

<sup>(2)</sup> Horowitz, G. *Adv. Mater.* **1998**, *10*, 365.

<sup>(3)</sup> Horowitz, G. *J. Mater. Chem.* **1999**, *9*, 2021.

explore new structures for high mobility, especially, the *π*-stacking structure.

Instead of aligning in the face-to-edge manner of the herringbone pattern, *π*-stacked molecules are situated in a face-to-face arrangement.<sup>22</sup> Clearly, such a packing configuration leads to stronger  $\pi-\pi$  interactions between neighboring molecules. The larger *<sup>π</sup>*-*<sup>π</sup>* interactions facilitate charge transfer between adjacent molecules by either of two mechanisms. At low temperatures, charge transport in crystalline materials appears to be coherent, i.e., band theory seems to be appropriate. At higher temperatures, phonon scattering leads to incoherent (hopping) transport.<sup>1,14,23</sup> Large  $\pi-\pi$  interactions increase the carrier bandwidth (and, hence, the mobility) at low temperatures, and decrease hopping barriers at higher temperatures. Calculations show that the bandwidth for *<sup>π</sup>*-stacking structures could reach 0.2- 0.4 eV, compared to the bandwidth of pentacene that is estimated to be  $0.2$  eV.<sup>23-25</sup> These values are still an order of magnitude smaller than those for inorganic semiconductors, e.g., Si, that have chemical bonds  $(1-3)$ eV) between neighboring atoms.

In this contribution, we develop several new materials for application in OTFTs. The materials are co-oligomers based on 4,4′-dialkyl-2,2′-bithiazole and 3,4-ethylenedioxythiophene (EDOT) in a donor-acceptor type molecule. Alkyl bithiazole oligomers crystallize with a "staircase" *π*-stack motif and may be reversibly n-doped,26-<sup>29</sup> whereas EDOT is a popular p-type material with low oxidation potential.30,31 When the two moieties are connected in a donor-acceptor system, the resultant materials will be both p- and n-dopable, and may adopt the *π*-stacked structures characteristic of the bithiazoles. Thus, we synthesized several co-oligomers: the acceptor-donor-acceptor type,  $AT_2A$  (A = 4,4'dialkyl-2,2'-bithiazole,  $T_2$  = EDOT dimer). Specific alkylbithiazoles will be denoted by  $E =$  diethylbithiazole,  $B =$  dibutylbithiazole, and  $H =$  dihexylbithiazole. Also synthesized were donor-acceptor-donor type oligomers, TBT ( $T = EDOT$ ,  $B =$  dibutylbithiazole) and  $T_2BT_2$  ( $T_2$  = EDOT dimer, B = dibutylbithiazole). Stille coupling reactions have been used successfully to prepare these compounds.<sup>32</sup> The full characterization of these materials has confirmed that the  $BT_2B$  oligomer is potentially a good candidate for high mobility organic semiconductors in OTFTs.

- (22) Hunter, C. A.; Sanders, J. K. M. *J. Am. Chem. Soc.* **1990**, *112*, 5525.
- (23) Pope, M.; Swenberg, C. E. *Electronic Processes in Organic Crystals and Polymers*, 2nd ed.; Oxford University Press: New York, 1999.
- (24) Silinsh, E. A.; Ca´pek, V. *Organic Molecular Crystals: Interaction, Localization and Transport Phenomena*; American Institute of Physics: Woodbury, NY, 1994.
- (25) Brédas, J. L.; Beljonne, D.; Cornil, J.; Calbert, J. P.; Shuai, Z.; Silbey, R. *Synth. Met.* **2001**, *125*, 107.
- (26) Koren, A. B.; Curtis, M. D.; Kampf, J. W. *Chem. Mater.* **2000**, *12*, 1519.
- (27) Curtis, M. D.; Cheng, H. T.; Nanos, J. I.; Nazri, G. A. *Macromolecules* **1998**, *31*, 205.
- (28) Nanos, J. I.; Kampf, J. W.; Curtis, M. D. *Chem. Mater.* **1995**, *7*, 2232.
- (29) Yamamoto, T.; Suganuma, H.; Maruyama, T.; Inoue, T.; Muramatsu, Y.; Arai, M.; Komarudin, D.; Ooba, N.; Tomaru, S.; Sasaki, S.; Kubota, K. *Chem. Mater.* **1997**, *9*, 1217.
- (30) Groenendaal, B. L.; Jonas, F.; Freitag, D.; Pielartzik, H.; Reynolds, J. R. *Adv. Mater.* **2000**, *12*, 481.
- (31) Sotzing, G. A.; Reynolds, J. R.; Steel, P. J. *Adv. Mater.* **1997**, *9*, 795.
- (32) Stille, J. K. *Angew. Chem. Int. Ed. Engl.* **1986**, *25*, 508.



## **Experimental Section**

**Materials.** All manipulations and preparations were performed under a nitrogen atmosphere using standard Schlenk line techniques unless otherwise stated. Reagents were purchased and used as received unless otherwise stated. Freshly opened, reagent grade solvents were dried by sequentially passing them through columns of molecular sieve (4A) and Cu/  $\text{Al}_2\text{O}_3$  under a dry  $\text{N}_2$  atmosphere. Synthesis of EDOT dimer (EDOT)2, <sup>31</sup> 4,4′-dialkyl-2,2′bithiazoles, 5,5′-dibromo-4,4′-dialkyl-2,2′-bithiazole, and 5-trimethylstannyl-4,4′-dialkyl-2,2′-bithiazole were performed according to the literature procedures.<sup>28</sup> Synthesis of  $ET_2E$  and  $HT_2H$  are similar to that of  $BT_2B$  and are described in the Supporting Information.

1H NMR spectra were collected on a Varian 300 or Varian 400 spectrometer and referenced to the residual solvent proton resonance. UV-vis spectra were collected on a Shimadzu 3101PC with baseline correction. Emission spectra were collected on a Shimadzu 4121 interfaced with a Gateway computer. Cyclic voltammetry (CV) was performed with a Princeton Applied Research Potentiostat model 263A interfaced to a PC computer. The solvent was dry  $CH<sub>3</sub>CN$  or THF, and the supporting electrolyte was tetrabutylammonium hexafluorophosphate (TBAPF<sub>6</sub>) (0.10 M). The reported potentials are vs the ferrocene/ferrocenium  $(Fc/Fc^{+})$  couple, obtained by adding a crystal of ferrocene to the solution. Mass spectra were collected on a VG 70-250-s high-resolution spectrometer. Elemental analyses were performed by the University of Michigan Microanalysis Laboratory.

**5,5**′**-Dibromo-2,2**′**-Bis(3,4-ethylenedioxythiophene) (Br**- **(EDOT)2**-**Br).** A 10-mL Schlenk flask equipped with a stirring bar was charged with  $(EDOT)_2$  (500 mg, 1.77 mmol) in dry  $CH_2Cl_2$  (60 mL). N-bromosuccinimide (NBS) (660 mg, 3.71 mmol) was added, and the reaction was allowed to stir at 0 °C for 2 h. The slightly blue mixture was dissolved in 500 mL of CH<sub>2</sub>Cl<sub>2</sub> and washed by NH<sub>3</sub>·H<sub>2</sub>O (10%, 150 mL), H<sub>2</sub>O (3  $\times$  50 mL), and brine ( $2 \times 50$  mL), and dried over MgSO<sub>4</sub>. The CH<sub>2</sub>-Cl2 was evaporated (Rotovap) to give a yellowish solid that was stored in a drybox. Yield = 660 mg (85%). <sup>1</sup>H NMR (CDCl<sub>3</sub>):  $\delta$  4.31 (8H,s).

**5,5**′**-Bis(4,4**′**-dibutyl-2,2**′**-bithiazol-5-yl)**-**(3,4,3**′**,4**′**-bis- (ethylenedioxy)-2,2**′**-dithienyl) (BT2B).** (Dibutylbithiazole)- SnMe3 (295 mg, 0.666 mmol) in dry toluene (60 mL) was added via cannula to a 100-mL Schlenk flask containing Br-  $(EDOT)_2-Br$  (130 mg, 0.30 mmol) and Pd(PPh<sub>3</sub>)<sub>2</sub>Cl<sub>2</sub> (21 mg, 0.03 mmol). The mixture was heated for 20 h during which time it turned dark red. After the mixture was cooled and filtered through Celite diatomaceous earth, the solvent was removed. The dark red solid was redissolved in  $CH_2Cl_2$  and washed with HCl (5 wt %, 50 mL), H<sub>2</sub>O (3  $\times$  50 mL), NH<sub>4</sub>Cl (saturated,  $2 \times 50$  mL), and brine ( $2 \times 50$  mL), and dried over MgSO4. The solvent was evaporated to give a red solid. Further purification involved a Soxhlet extraction with methanol to remove most of the starting materials, followed by flash chromatography over silica gel with  $CH_2Cl_2$  eluent to remove side-products. Bright red, single crystals were grown by slowly evaporating the solvent. Yield = 180 mg (71%). <sup>1</sup>H NMR (CDCl<sub>3</sub>):  $\delta$  0.915 (6H, t, *J* = 7.2 Hz, -CH<sub>3</sub>, outside),  $\delta$  0.95 (6H, t,  $J = 7.2$  Hz,  $-CH_3$ , inner),  $\delta$  1.37 (4H, m,  $J = 7.6$  Hz,  $-CH<sub>2</sub>$ , outside), *δ* 1.44 (4H, m,  $-CH<sub>2</sub>$ , inner), *δ* 1.69 (4H, m, −CH<sub>2</sub>−, outside),  $\delta$  1.75 (4H, m, −CH<sub>2</sub>−, inner),  $\delta$  2.78 (4H, t,  $J = 8$  Hz,  $-CH_2$ , outside),  $\delta$  2.98 (4H, m,  $J = 7.6$  Hz, -CH<sub>2</sub>-, inner), *δ* 4.36 (8H, t, *J* = 2 Hz, -CH<sub>2</sub>CH<sub>2</sub>-, EDOT ring),  $\delta$  6.90 (2H, s, 5-position ring proton). MS (EI):  $M^+$  = 838.3. Elemental Analysis: calcd C, 57.3%; H, 5.53%; N, 6.67%; found C, 57.1%; H, 5.63%; N, 6.61%.

**2-(Trimethylstannyl)-3,4-(ethylenedioxy)thiophene.** In a 100-mL Schlenk flask, was placed a 2.0 g (14.1 mmol) sample of 3,4-(ethylenedioxy)thiophene in 50 mL of dry THF. A solution (6.2 mL, 15.5 mmol) of 2.5 M *n*-butyllithium was added dropwise into the EDOT solution at  $-78$  °C under nitrogen. After the solution was stirred for 1 h, it was allowed to warm to  $-40$  °C, and 3.37 g (16.9 mmol) of trimethylstannyl chloride was added to the solution. The solvent was removed via Rotovap after the solution was stirred overnight. After the solvent was evaporated, the residue was dissolved in hexane and washed with H<sub>2</sub>O (3  $\times$  30 mL), then brine (2  $\times$  30 mL), and dried over MgSO<sub>4</sub>. The product was a white solid after the solvent was evaporated in a vacuum. Yield =  $3.60$  g (84%). <sup>1</sup>H NMR (CDCl<sub>3</sub>): *δ* 0.31 (9H, s,  $-SnMe_3$ ), *δ* 4.14 (4H, t, *J* = 8.0 Hz, -CH2- on EDOT ring), *<sup>δ</sup>* 6.55 (1H, s, 5-position ring proton). Elemental analysis: calcd C, 35.5%; H, 4.63%; found C, 34.7%; H, 4.54%.

**5,5**′**-Bis(3,4-ethylenedioxythien-2-yl)-4,4**′**-dibutyl-2,2**′ **bithiazole (TBT).** In a 100-mL Schlenk flask equipped with a reflux condenser and nitrogen outlet, was placed (EDOT)- SnMe3 (146 mg, 0.479 mmol), Br-(Dibutylbithiazole)-Br (100 mg, 0.228 mmol), and Pd(PPh3)2Cl2 (16 mg, 0.02 mmol), and dry toluene (60 mL). The mixture was heated to reflux for 16 h, after which time the solution showed a green fluorescence. After cooling to room temperature, the toluene solution was washed with HCl solution (5%, 50 mL), H<sub>2</sub>O (3  $\times$  30 mL), saturated NH<sub>4</sub>Cl solution (2  $\times$  30 mL), then brine (2  $\times$  30 mL), and dried over MgSO4. The solvent was removed by Rotovap to give an orange solid. Then, the raw product in  $CH_2Cl_2$  was run through a chromatography column filled with silica gel for further purification. Yield = 109 mg (85%). <sup>1</sup>H NMR  $(CDCI_3)$ :  $\delta$  0.944 (6H, t,  $J = 7.2$  Hz,  $-CH_3$ ),  $\delta$  1.44 (4H, m, J  $= 7.2$  Hz,  $-CH_2$ , $\delta$  1.75 (4H, m,  $J = 7.2$  Hz,  $-CH_2$ ),  $\delta$  2.95 (4H, t,  $-CH_2$ ),  $\delta$  4.26 (4H, m,  $J = 2.7$  Hz,  $-CH_2$  on EDOT ring),  $\delta$  4.32 (4H, m,  $J = 2.7$  Hz,  $-CH_2$  on EDOT ring),  $\delta$  6.41 (2H, s, EDOT ring proton at 5-position). Elemental analysis: calcd C, 55.7%; H, 5.03%; N, 4.99%; found C, 55.8%; H, 5.09%; N, 4.85%.

**5,5**′**-Bis(5-bromo-3,4-ethylenedioxythien-2-yl)-4,4**′**-dibutyl-2,2**′**-bithiazole (Br**-**TBT**-**Br).** A 100-mL Schlenk flask equipped with a stirring bar was charged with TBT (100 mg,  $0.178$  mmol) in  $CH_2Cl_2$  (60 mL). N-bromosuccinimide (NBS) (67 mg, 0.376 mmol) was added, and the reaction was allowed to stir at 0 °C and under exclusion of light for 2 h. After the solvent was evaporated, the raw product was washed with acetone to remove the extra starting materials to give an orange solid. Yield  $= 121$  mg (95%). <sup>1</sup>H NMR (CDCl<sub>3</sub>):  $\delta$  0.94  $(6H, t, J = 7.2 \text{ Hz}, -CH_3), \delta$  1.25 (4H, m,  $J = 7.2 \text{ Hz}, -CH_2-$ ), *δ* 1.75 (4H, m,  $J = 7.2$  Hz,  $-CH_2$ ), *δ* 2.95 (4H, t,  $-CH_2$ ), *δ* 4.32 (8H, m,  $-CH_2$ , EDOT ring). Elemental analysis: calcd C, 43.5%; H, 3.65%; N, 3.90%; found C, 45.3%; H, 4.16%; N, 4.04%.

**5,5**′**-Bis(3,4,3**′**,4**′**-bis(ethylenedioxy)-2,2**′**-dithien-5-yl)- 4,4**′**-dibutyl-2,2**′**-bithiazole (T2BT2).** In a 100-mL Schlenk flask equipped with a reflux condenser and nitrogen outlet, was placed (EDOT)-SnMe<sub>3</sub> (94 mg, 0.308 mmol), Br-TBT-Br (100 mg, 0.139 mmol), Pd(PPh<sub>3</sub>)<sub>2</sub>Cl<sub>2</sub> (20 mg, 0.03 mmol), and dry toluene (60 mL). The mixture was heated to reflux for 12 h. After cooling to room temperature, the solution was washed with HCl solution (5%, 50 mL),  $H_2O$  (3  $\times$  30 mL), saturated NH<sub>4</sub>Cl solution ( $2 \times 30$  mL), then brine ( $2 \times 30$  mL),

Scheme 1. Synthesis of AT<sub>2</sub>A Oligomers: (a) **BuLi/-78** °**C/THF, CuCl<sub>2</sub>; (b) NBS/CH<sub>2</sub>Cl<sub>2</sub>/0** °**C**; (c) **BuLi/**-**<sup>78</sup>** °**C/THF, SnMe3Cl; (d) Stille Coupling, Pd(PPh3)2Cl2/Toluene/20 h**



and dried over MgSO4. The solvent was removed by Rotovap to give an orange solid. Then, the raw product in  $CH_2Cl_2$  was run through a chromatography column filled with silica gel for further purification. Yield = 109 mg (93%). <sup>1</sup>H NMR (CDCl<sub>3</sub>):  $\delta$  0.976 (6H, t,  $J = 7.2$  Hz,  $-C\overline{H}_3$ ),  $\delta$  1.45 (4H, m,  $J = 7.2$  Hz,  $-CH_2$ -),  $\delta$  1.79 (4H, m,  $J = 7.2$  Hz,  $-CH_2$ -),  $\delta$ 3.00 (4H, t,  $J = 7.8$  Hz,  $-CH_2$ ),  $\delta$  4.25-4.39 (16H, m,  $-CH_2$ , EDOT ring), *δ* 6.32 (2H, s, EDOT ring proton at 2-position). Elemental analysis: calcd C, 54.3%; H, 4.31%; N, 3.33%; found C, 53.2%; H, 4.65%; Nn 3.18%.

## **Results and Discussion**

**Preparation.** All these co-oligomers of alky-bithiazoles and EDOT were prepared using Stille coupling reactions.  $AT_2A$  oligomers were synthesized as shown in Scheme 1. The coupling between the diallkylbithiazole-SnMe<sub>3</sub> and Br- $(EDOT)_2$ -Br required a high reaction temperature because the bithiazole-tin reagents are poor nucleophiles. Thus, toluene was chosen as the solvent in most of our coupling reactions, as reactions in THF gave very low yields (∼20%). The yields of the coupling reactions are reasonably high, except for  $ET_2E$ . It is necessary to have pure reactants to achieve high yields. Using less than 1 equiv of BuLi per bithiazole in the lithiation reaction decreased the amount of ditin bithiazole formed by unintentional dilithiation. Furthermore, because of the low oxidation potential of (EDOT)<sub>2</sub>, Br-(EDOT)<sub>2</sub>-Br must be prepared at 0 °C with exclusion of light to prevent over-oxidation or polymerization by NBS. Reducing the crude product and removal of the NBS with NH3'H2O is recommended. However, the  $AT_2A$  oligomers are reasonably stable due to the electron-withdrawing bithiazole groups attached to the  $(EDOT)_2$  core that block the reactive 5,5'-positions of EDOT2. The products with butyl and hexyl sidechains are very soluble in many common solvents, e.g., toluene, and  $CHCl<sub>3</sub>$ , making them suitable for liquidphase spin-coating.

The syntheses of TBT and  $T_2BT_2$  are shown in Scheme 2. The Stille coupling reactions in this series gave better yields than those obtained for  $BT_2B$  because (EDOT)-SnMe3 is a better nucleophile than (dibutylbithiazole) $-SnMe<sub>3</sub>$ . Adding a slight excess of (EDOT) $-$ SnMe<sub>3</sub> reduced the amount of monosubstituted product that formed during the coupling reaction.







**Figure 1.** WAXD patterns of solution-cast films of  $ET_2E$ ,  $BT_2B$ , and  $HT_2H$ .

The only successful route to  $T_2BT_2$  was the coupling between (EDOT)-SnMe<sub>3</sub> and Br-TBT-Br. Attempts at coupling  $(EDOT)_2$ -SnMe<sub>3</sub> and Br-(dibutylbithiazole)-Br failed because of the low stability of  $(EDOT)_2$ -SnMe3. Fortunately, Br-TBT-Br was easy to prepare by bromination of TBT by NBS. Because Br-TBT-Br was only partially soluble in toluene, the coupling reaction had to be done at reflux temperature. Both compounds, TBT and  $T_2BT_2$ , are quite soluble and stable enough for further characterization.

**X-ray Diffraction.** Since the crystal structure of  $BT_2B$  has been solved recently,<sup>33</sup> the film orientation could be studied based on the wide-angle X-ray diffraction (WAXD) patterns. Figure 1 shows an overlay of the WAXD patterns of AT2A solution-cast films. The 2*θ* and *d* values are collected in Table 1.  $BT_2B$  shows XRD peaks at  $2\theta = 6.3^{\circ}$ , 10.9°, 18.9°, 22.5°, and 25.1°, corresponding to 14.0, 8.1, 4.7, 3.9, and 3.5 Å, respectively. These peaks could be assigned to the  $(101)$ ,  $(002)$ ,  $(113)$ ,  $(311)$ , and  $(312)$  reflections by comparison to the X-ray powder pattern simulated from the single-crystal structure (Supporting Information). The *d* spacing corresponding to the peak at  $2\theta = 25.1^{\circ}$  is similar to the *π*-stacking distance measured from the single-crystal structure.

There is also a strong diffraction peak at  $2\theta = 5.0^{\circ}$ from  $HT<sub>2</sub>H$  films, corresponding to the lamellar struc**Scheme 2. Synthesis of TBT and T2BT2 Oligomers: (a) Br2/CHCl3/Reflux/2 h; (b) BuLi/**-**<sup>78</sup>** °**C/THF, SnMe<sub>3</sub>Cl; (c) Stille Coupling, Pd(PPh<sub>3</sub>)<sub>2</sub>Cl<sub>2</sub>/Toluene/ 16 h; (d) NBS/CH2Cl2/0** °**C/Dark; (e) Stille Coupling, Pd(PPh3)2Cl2/Toluene/16 h**



ture with a *d* spacing of 17.7 Å. The longer hexyl sidechain is expected to expand the lamellar distance. Other peaks of HT2H are located at 2*θ* of 9.8°, 17.5°, 21.9°, and 25.2°, and these match the *d* spacings of 9.0, 5.1, 4.1, and 3.5 Å in the butyl derivative. It thus appears that the butyl and hexyl derivatives are isomorphous.

The structure of  $ET_2E$  appears to be similar, but not isomorphous, to those of the butyl and hexyl compounds. A strong peak at  $2\theta = 4.4^{\circ}$  (20 Å) is observed, but the 20 Å distance is not compatible with a simple lamellar spacing between ethyl-substituted oligomers. Other peaks are located at 2*θ* of 8.1°, 8.5°, 9.7°, 12.7°, 17.1°, 21.2°, and 25.1°, corresponding to 10.9, 10.4, 9.1 7.0, 5.2, 4.2, and 3.5 Å. However, the peak corresponding to the *π*-stacking distance (3.5 Å) still can be observed, implying that  $ET_2E$  also forms  $\pi$ -stacks.

**UV**-**Vis Spectroscopy.** Figures 2 and 3 display the (33) Cao, J.; Kampf, J. W.; Curtis, M. D. **2002**, manuscript in  $UV - vis$  **Spectroscopy.** Figures  $\lambda$  and 3 display the exparation.

preparation.



**Figure 2.** UV-vis absorption spectra of  $AT_2A$ , TBT, and  $T_2$ - $BT_2$  oligomers in solution: (a) TBT, (b)  $T_2BT_2$ , (c)  $ET_2E$ , (d)  $BT_2B$ , and (e)  $HT_2H$ .



**Figure 3.** UV-vis absorption spectra of the oligomer films. Top: (a)  $ET_2E$ , (b)  $BT_2B$ , and (c)  $HT_2H$ . Bottom: (a) TBT and (b)  $T_2BT_2$ .

oligomers, and the results are summarized in Table 1. As expected, the length of alkyl side-chains slightly influenced the absorption spectra of  $AT_2A$  compounds in both solution and the solid state. For example,  $ET_2E$ in solution gave an absorption maximum at 2.69 eV (461 nm), which is 0.03 eV blue-shifted compared to  $BT_2B$ and  $HT_2H$ , both of which absorb at 2.66 eV (466 nm). Interestingly, if we compare two oligomers having the identical side-chains,  $T_2BT_2$  and  $BT_2B$ , there is no difference in solution UV absorption maxima as both occur at 2.66 eV. The donor-acceptor sequence seems to have little influence on the solution absorption, as long as the number of the conjugated rings is the same. Compared to the corresponding donor or acceptor oligomers, the donor-acceptor systems have clearly lowered the energy of the optical transition. For instance, butyl bithiazole trimer  $(B_3)$ , a typical acceptor oligomer,

Table 2. Estimates of Davydov Splitting ( $W<sub>D</sub>$ ) and **Interchain Transfer Integral (***â***) from the Solid-State Absorption Spectra and Solution Emission Spectra***<sup>a</sup>*

name	$W_D$ (eV)	$\Delta E_{\rm abs}$ (eV)	$\Delta E_{\rm em}$ (eV)	$\beta$ (meV)
ET,E	0.21	1.34	0.75	95
BT <sub>2</sub> B	0.15	1.08	0.58	88
HT <sub>2</sub> H	0.16	1.10	0.60	85
<b>TBT</b>	0.17	1.06	0.75	35
$T_2BT_2$	0.15	1.10	0.73	55

*<sup>a</sup>* ∆*E*abs and ∆*E*em represent the measured widths of the absorption and emission peaks, respectively.

shows an absorption maximum at 3.27 eV (379 nm), compared to 2.66 eV for  $BT_2B$ . Because of the fewer conjugated rings in its backbone, a significant blue shift was observed in the absorption maximum of TBT at 3.02 eV (410 nm) compared to 2.66 eV (466 nm) for  $BT_2B$ .

The UV spectra in solid films are dependent on the solid-state morphology, in particular, whether the molecules are  $\pi$ -stacked.<sup>34</sup> In the series of AT<sub>2</sub>A, the absorption peaks are all shifted to lower energies with respect to the solution because molecular chains adopt a more planar conformation in the solid state.  $ET_2E$ ,  $BT_2B$ , and  $HT_2H$  share a 3-peak pattern with maxima located at 2.69, 2.48, and 2.25 eV (461, 500, and 551 nm, respectively) for  $ET_2E$ , 2.61, 2.48, and 2.30 eV (475, 500, and 539 nm, respectively) for  $BT_2B$ , and 2.66, 2.50, and 2.30 eV for  $HT<sub>2</sub>H$  (466, 496, and 539 nm, respectively). The compounds, TBT and  $T_2BT_2$ , also have a 3-peak pattern similar to that of  $BT_2B$ . The peak maxima are situated at 3.03, 2.86, and 2.63 eV (409, 434, and 471 nm, respectively) for TBT and 2.65, 2.50, and 2.30 eV (468, 496, and 539 nm, respectively) for  $T_{2}$ - $BT_2$ . The origin of the multi-peaks has been explained by exciton band theory.<sup>26,34-39</sup> If there is more than one molecule per unit cell, the absorption peaks are split into one allowed transition for each translationally inequivalent molecule (Davydov splitting). Phonon coupling may relax the  $\Delta k = 0$  selection rule ( $k =$  wavevector for the exciton band) with the result that the absorption peak samples the density of states in the exciton band. This, combined with the Davydov splitting  $(W<sub>D</sub>)$ , results in a 3-band pattern if there are two molecules per unit cell.39

The Davydov splitting can be estimated from the film absorption spectra, i.e. the separation of the peaks from the two high-energy exciton branches.<sup>26</sup> For example, in the case of  $BT_2B$ , the Davydov splitting is about 0.15 eV ( $W_D = 2.61 - 2.46 = 0.15$  eV). The values of the five co-oligomers are in the range of  $0.15-0.21$  eV, as collected in Table 2, which are larger than those normally found in organic crystals such as anthracene. Our recent study shows that the interchain transfer integral,  $\beta$ , can also be estimated from the solid-state

<sup>(34)</sup> Politis, J. K.; Nemes, J. C.; Curtis, M. D. *J. Am. Chem. Soc.* **2001**, *123*, 2537.

<sup>(35)</sup> Beljonne, D.; Langeveld-Voss, B. M. W.; Shuai, Z.; Janssen, R. A. J.; Meskers, S. C. J.; Meijer, E. W.; Bredas, J. L. *Synth. Met.* **1999**, *102*, 912.

<sup>(36)</sup> Bosisio, R.; Botta, C.; Colombo, A.; Destri, S.; Porzio, W.; Grilli, E.; Tubino, R.; Bongiovanni, G.; Mura, A.; DiSilvestro, G. *Synth. Met.* **1997**, *87*, 23.

<sup>(37)</sup> DiCesare, N.; Belletete, M.; Marrano, C.; Leclerc, M.; Durocher, G. *J. Phys. Chem. A* **1999**, *103*, 795.

<sup>(38)</sup> Spano, F. C.; Siddiqui, S. *Chem. Phys. Lett.* **1999**, *314*, 481. (39) Koren, A. B.; Kampf, J.; Curtis, M. D. **2002**, submitted for publication.

**Table 3. Cyclic Voltammetry of Five Oligomers as Measured in THF/TBAPF6 Solution; Potentials are Referenced to Fc/Fc**<sup>+</sup> **Couple**

		oxidation (V vs $Fc/Fc^+$ )			reduction (V vs $Fc/Fc^+$ )		
name	$E_{\alpha}P$	$E^{1/2}$ <sub>ox</sub>	$\Delta E_{p}$	$E_{\rm red}P$	$E^{1/2}$ red	$\Delta E_{\rm n}$	$E_{\rm g}^{\rm CV}$
$ET_2E$	0.282 0.403			$-2.45$	$-2.37$	0.16	2.7
$BT_2B$	0.302 0.761	0.262 0.724	0.080 0.075	$-2.38$	$-2.31$	0.15	2.6
HT <sub>2</sub> H	0.316 0.656	0.281 0.593	0.071 0.13	$-2.37$	$-2.32$	0.11	2.6
<b>TBT</b>	0.637			$-2.42$	$-2.31$	0.22	2.9
$T_2BT_2$	0.130 0.220			$-2.35$	$-2.26$	0.18	2.4

absorption and emission spectra which are discussed elsewhere.39 The value of 4*â* can be obtained from eq 1

$$
4\beta = W_{\rm abs} - \Delta E_{\rm em} \tag{1}
$$

where *W*abs and ∆*E*em correspond to the width of the solid-state absorption and emission spectra, respectively. The effect of the Davydov splitting has been subtracted from the measured width of absorption spectra:  $W_{\text{abs}} = \Delta E_{\text{abs}} - W_{\text{D}}$ . The values of  $\beta$ , as listed in Table 2, imply that  $AT_2A$  oligomers ( $\beta \approx 90$  meV) may have stronger interchain interactions than TBT or T<sub>2</sub>- $BT_2$  ( $\beta = 35$  or 55 meV) and, in the donor-acceptordonor series, *â* increases as the chain length increases. We have used the ZINDO method to estimate the interchain interactions of *π*-stacks, and the magnitudes of the calculated transfer integrals are comparable to those found experimantally.<sup>40</sup> It is interesting to note that the fine structures seen in  $ET_2E$ ,  $BT_2B$ , and  $HT_2H$ film absorption spectra imply that the molecules are *π*-stacked, and that there are two molecules per unit cell-features that are confirmed by the solved crystal structure of  $BT_2B$ .<sup>33</sup> Therefore, we may also predict that the solid-state structures of TBT and  $T_2BT_2$  feature molecules that are *π*-stacked with two inequivalent molecules per unit cell.

Emission spectra of solutions of the co-oligomers  $AT_2A$ are shown in Figure 4, and the results are also summarized in Table 1.  $ET<sub>2</sub>E$  has an emission maximum at 2.23 eV (556 nm). The maxima for  $BT_2B$  and  $HT_2H$ are nearly identical at 2.12 eV (585 nm), and the spectrum of  $BT_2B$  also displays a small shoulder near 2.20 eV (564 nm). As expected, the high-energy edges of the emission from all three co-oligomers overlap the low-energy feet of their corresponding absorption peaks (overlap of the 0-0 vibronic transitions). However, in the cases of TBT and  $T_2BT_2$  oligomers, multiple peaks were detected in emission spectra. TBT has maxima at 2.57 and 2.45 eV (482 and 506 nm, respectively), whereas  $T_2BT_2$  emits at somewhat lower energy values: 2.24 eV (554 nm) and 2.12 eV (585 nm). Compared to their absorption spectra, the high-energy edges of the emission also overlap the low-energy feet of their corresponding absorption peaks, but the Stokes shifts are larger than those of  $AT<sub>2</sub>A$  compounds, implying a large degree of out-of-plane ring torsion in the ground state of TBT and  $T_2BT_2$ .

The band gaps of these co-oligomers were estimated by extrapolation of the low energy edge of the absorption spectra to the baseline and are also collected in Table



**Figure 4.** Emission spectra of the oligomers in dilute CHCl<sub>3</sub> solution. Top: (a)  $ET_2E$ , (b)  $BT_2B$ , and (c)  $HT_2H$ . Bottom: (a) TBT and (b)  $T_2BT_2$ .

1. Both  $BT_2B$  and  $HT_2H$  have the same band-gaps (2.3) and 2.1 eV for the solution and solid state, respectively), whereas  $ET_2E$  gives a little smaller value (2.28 and 2.00 eV for the solution and solid-state, respectively). In the case of  $T_2BT_2$ , the values are very similar to  $BT_2B$  and HT<sub>2</sub>H (2.24 and 2.10 eV for the solution and solid-state, respectively). As expected, the smallest oligomer, TBT, has the highest band gap (2.60 and 2.46 eV for the solution and solid-state, respectively) due to its shorter conjugation length.

**Electrochemistry.** Cyclic voltammetry (CV) measurements were carried out to determine the redox behavior of the oligomers, and the results are shown in Table 3. The oxidation and reduction waves of the  $AT_2A$ materials in 0.1 M THF/TBAPF $_6$  solutions are displayed in Figures 5 and 6. As expected, the acceptor-donoracceptor type oligomers,  $AT_2A$ , can be both oxidized and reduced within the potential range of the solvent/ electrolyte window. In solution, all three  $AT_2A$  oligomers have two oxidation waves.  $HT<sub>2</sub>H$  has one reversible (40) Cao, J.; Curtis, M. D. **2002**, manuscript in preparation. oxidation near  $E^{1/2}{}_{ox} = 0.281$  V ( $\Delta E_p = 71$  mV) and a



**Figure 5.** Oxidation cyclic voltammetry waves of the oligomers at 50 mV/s scan rate in THF/TBAPF $_6$  solution. Top: three  $AT_2A$  oligomers. Bottom: TBT and  $T_2BT_2$ .

quasi-reversible oxidation at E  $^{1/2}$ <sub>ox</sub> = 0.593 V ( $\Delta E$ <sub>p</sub> = 130 mV) vs the ferrocene/ferrocenium couple  $(Fc^+/Fc)$ , 0.1 M Bu<sub>4</sub>NPF<sub>6</sub> (TBAPF<sub>6</sub>) in THF as the supporting electrolyte). Similarly, BT2B also shows two quasireversible oxidations around  $E^{1/2}$ <sub>ox</sub> = 0.262 V ( $\Delta E$ <sub>p</sub> = 80 mV) and  $E^{1/2}_{\text{ox}} = 0.724$  V ( $\Delta E_{\text{ox}} = 75$  mV). These<br>two quasi-reversible couples correspond to 1e-ovidations two quasi-reversible couples correspond to 1e-oxidations to form cation radical and di-cations, respectively. The potential could be cycled repeatedly with no change in the CV curves. However, the oxidation behavior of  $ET_2E$ is quite different from that of  $BT_2B$  or  $HT_2H$ .  $ET_2E$  has two resolvable oxidation waves at  $E_{\text{ox}} = 0.282$  and 0.403<br>V and one returning reduction wave at 0.113 V. The V and one returning reduction wave at 0.113 V. The oxidation waves of  $ET_2E$  drift to more positive values by almost 60 mV after being cycled for 10 times, instead of being invariant as in  $HT_2H$  and  $BT_2B$ . We ascribe these changes to the formation of a film of the oxidized material on the electrode surface. The longer alkyl side chains apparently inhibit film formation. Despite the variation in the oxidation behavior, the reduction cycles are very similar among the three compounds.  $HT_2H$ displays a quasi-reversible reduction near  $E^{1/2}$ <sub>red</sub> =  $-2.32$  V ( $\Delta E_{\text{red}} = 110$  mV). BT<sub>2</sub>B shows an almost identical reduction potential at  $F^{1/2}$ <sub>m</sub> = -2.31 V ( $\Delta E_{\text{red}}$ ) identical reduction potential at  $E^{1/2}$ <sub>red</sub> = −2.31 V ( $\Delta E$ <sub>red</sub> = 150 mV) and  $E$ <sup>T</sup><sub>0</sub></sub>F displays very little difference  $= 150$  mV), and ET<sub>2</sub>E displays very little difference, with a reduction potential at  $E^{1/2}{}_{\text{red}} = -2.37 \text{ V}$  (∆*E*<sup>p</sup><sub>red</sub> = 160 mV)  $= 160$  mV).

Compared with the acceptor-donor-acceptor type of structures, the donor-acceptor-donor compounds, TBT and  $T_2BT_2$ , show quite different oxidation behavior. Both compounds show electrochemical irreversibility during the oxidation cycle. TBT gave an oxidation at  $E_{ox} = 0.637$  V and a returning reduction at  $E_{red} = 0.347$ 



**Figure 6.** Reduction cyclic voltammetry waves of the oligomers at 50 mV/s scan rate in THF/TBAPF6 solution. Top: three  $AT_2A$  oligomers. Bottom: TBT and  $T_2BT_2$ .

**Scheme 3. Energy Diagram of the Co-oligomers, Based on Their Threshold Oxidation and Reduction Potentials and the Low-energy UV**-**Vis Absorption Thresholds**



eV, whereas  $T_2BT_2$  had two oxidations at 0.130 and 0.220 V with a returning  $E_{\text{red}}$  at  $-0.12$  V. A more detailed study of the spectroelectrochemical and chemical oxidations of these compounds will be reported separately.<sup>40</sup> Both TBT and  $T_2BT_2$  had similar quasireversible reduction potentials at  $E^{1/2}$ <sub>red</sub> = -2.31 V  $(\Delta E_{\text{red}} = 220 \text{ mV})$  and  $E_{\text{red}}^{1/2} = -2.26 \text{ V}$  ( $\Delta E_{\text{red}} = 180 \text{ mV}$ ) respectively Regardless of the donor-acceptor mV), respectively. Regardless of the donor-acceptor sequence of these co-oligomers, EHMO calculations suggest that the positive charge is mostly located on the EDOT rings after oxidation of the co-oligomers, a result that is confirmed by ESR experiments.40

The band gaps determined from CV are similar to the optical band gaps obtained from extrapolation of the low-energy edge of the UV-vis absorption. An approximate energy diagram, illustrated in Scheme 3, is based on the CV and UV-vis data. The band gaps calculated from CV are 2.6, 3.0, and 2.4 eV, compared



Figure 7. Spectroelectrochemical measurement of BT<sub>2</sub>B films: (a) neutral film, (b)  $0.60$  V for 1 min, (c)  $0.0$  V for 5 min, and (d) 0.0 V for 10 min.



**Figure 8.** I<sub>2</sub> doping of  $BT_2B$  films made by vacuum deposition: (a) neutral film, (b) doped by  $I_2$  vapor for 3 h.

with the corresponding optical band gaps: 2.3, 2.6, and 2.2 eV for  $AT_2A$ , TBT, and  $T_2BT_2$ , respectively.

A spectroelectrochemical study of the oligomer films was undertaken, and the results for  $BT_2B$  are shown in Figure 7. A film of  $BT_2B$  was dip-coated onto an ITO electrode and the measurements were conducted in a cell containing a  $0.1$  M TBAPF $_6$  solution in acetonitrile. The spectra of the films changed rapidly after voltage was applied. At 0.60 V, the intensities of the peaks of the neutral film decreased, while new peaks grew in at 2.4, 1.6, and 1.1 eV (517, 775, and 1127 nm, respectively). Upon resetting the voltage to 0.0 V, the film starts to de-dope as evidenced by the decrease in the intensities of the peaks assigned to the oxidized species and the concomitant growth of the peaks due to the neutral material. The fact that the fine structure, caused by the details of molecular packing, is regenerated also is strong evidence that the cycle of oxidation/reduction does not alter the solid-state packing. This means that the dopant ions  $(PF_6^-)$  most likely move into the channels that are parallel to the *π*-stacks. The de-doping process is relatively slow: it takes about 10 minutes after switching the voltage back to 0.0 V for the film to be fully reduced. After a doping/dedoping cycle, the intensity of the absorption peak is a little lower than that of the fresh sample, most likely because a small amount of oxidized film detaches from the ITO electrode in acetonitrile solution and is not reduced.

Solid-state films can also be doped by chemical means, e.g.,  $I_2$  vapor. Figure 8 displays the UV-NIR spectra of  $BT_2B$  film made by vapor deposition before and after  $I_2$ doping. Compared to the spectroelectrochemical changes observed in Figure 7, the chemical doping causes a similar spectral change. The peaks of the neutral oligomer are replaced by peaks at 2.48 and 1.01 eV (500 and 1228 nm) along with broad shoulders at 2.03 and 1.61 eV (611 and 770 nm). The explanation of the spectral changes in the oxidized films in terms of configuration interaction of the excited states of the radical cations has recently been made.25

The conductivities of the  $I_2$ -doped  $BT_2B$  films (about 1 *µ*m to 5 *µ*m thick) were measured. The films were cast from CHCl<sub>3</sub> onto glass slides and doped by  $I_2$  vapor for 3 h. The doped films gave conductivities of about 1  $\times$  $10^{-2}$  S/cm<sup>-1</sup> as measured by the standard 4-point probe method. The conductivity is lower than that of EDOT polymer,30 possibly due to the roughness of the solution cast films that introduces high grain-boundary resistance.

#### **Conclusions**

Alkyl-bithiaozle and  $(EDOT)_2$  co-oligomers,  $AT_2A$ , have been synthesized by Stille coupling reactions. The addition of alkyl-bithiazole units induces the co-oligomers to form *π*-stacked crystal structures with short intermolecular contact distances (e.g., 3.5 Å measured for the  $BT_2B$  crystal) and good  $\pi-\pi$  overlap. Analysis of the X-ray diffraction pattern of the films deposited on a hydrophilic substrate shows that the *π*-stacking axis is parallel to the substrate with the conjugated rings tilted with respect to the substrate. The incorporation of the EDOT dimer unit into the oligomers imparts a low oxidation potential. Because the bithiazole groups have a reasonably low reduction potential, the combination of low energy LUMO acceptor (bithiazole) with high energy HOMO donor (EDOT) leads to a lowered band gap. The electrochemical characterization demonstrates that either p-doping or n-doping is possible for these oligomers.  $BT_2B$  and  $HT_2H$  oligomers display quasireversible electrochemical behavior, whereas  $ET_2E$ , ATA, and  $T_2BT_2$  oligomers show more complex behavior. All these features make these new materials promising for use in organic-based TFTs. The fabrication and characterization of OTFTs with  $BT_2B$  as the active semiconductors will be reported elsewhere.

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**Supporting Information Available:** General experimental and synthetic procedures and crystallographic data tables of BT2B. This material is available free of charge via the Internet at http://pubs.acs.org.

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